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POSTER

# Vitaliy Gerasimov · Vladimir Mitsa · Yuriy Babinets Optical tomography of non-crystalline films by interference enhanced Raman spectroscopy

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Abstract The depth dependence of Raman spectra of a-GeS<sub>2</sub>-type films having a different optical thickness ( $\lambda/4$  and  $\lambda/2$ ) and their refractive index profile have been investigated. The model of a layered-inhomogeneous structure of films has been proposed. There have been distinguished three regions: near-surface region (up to 50 Å), central part and transition film-substrate region (up to 300 Å).

### Introduction

To overcome obstacles connected with the insufficient intensity of the Raman signal for structural investigations of thin and superthin films, the principles of adaptive optics have been used [1,2]. Their use for interference enhancement of Raman signal (IERS) allows one to redistribute the field intensity of an exciting light wave ( $\overline{E}^2$ ) inside a three-layered interference structure. The possibility to redistribute  $\overline{E}^2$  in the film is especially valuable while studying depth dependences of physical properties of thin films by optical methods within optical tomography [2]. The aim of the present work is to develop an IERS method to investigate depth dependences of structural peculiarities for thermally evaporated thin films of strongly dissociative a-GeS<sub>2</sub>-type glasses.

# Experimental

The a-GeS<sub>2</sub>-SiO<sub>2</sub>-Al three-layered structure was prepared by flash evaporation of GeS<sub>2</sub> powder onto a substrate with preliminarily deposited Al and SiO<sub>2</sub> (1600 Å) by conventional methods. The thickness of a-GeS<sub>2</sub>-type films during deposition was controlled by

V. Gerasimov · V. Mitsa · Y. Babinets Physical Department, Uzhgorod State University, Uzhgorod, 294000, Ukraine photometric methods. Raman spectra were taken with a DFS-24 instrument with a microoptical attachment in the geometry of back scattering of reflection. As the exciting source for the Raman signal a set of lasers with wavelengths of 4800, 6300 and 7500 Å was used. The distribution of the refractive index profile in films was studied by multiangular ellipsometry.

The distribution of the field in a light wave in the upper high-refractive layer of an a-GeS<sub>2</sub>-SiO<sub>2</sub>-Al structure was calculated using the equation given in [3].

#### **Results and discussion**

The distribution of the field in a-GeS<sub>2</sub> films with an optical depth of  $\lambda/4$  and  $\lambda/2$  is given in Fig. 1. The analysis of IERS for the near-surface region of As<sub>3</sub>(GeS<sub>2</sub>)<sub>97</sub> films multiple to  $\lambda/4$  and  $\lambda/2$  [3]  $(\lambda_{ex} = 4800; 7500 \text{ Å})$  and the data of the present investigations (Fig. 2) show that the position of bands in the region of  $200-400 \text{ cm}^{-1}$  is close to the position of bands in Raman spectra of  $Ge_xSe_{1-x}$  glasses rich in germanium  $(x = 33-40 \text{ at}\%, n = 2.5-2.6 \text{ at } \lambda = 6300 \text{ Å})$ . The band at  $335 \text{ cm}^{-1}$  (Fig. 2) can be related to a position close to the  $v_1(A)$  band in a-GeS<sub>2</sub>, while applying hydrostatic pressure [4], the band at  $290 \text{ cm}^{-1}$  coincides with the position of the band in the Raman spectrum of a-Ge [1]. The shift of the  $v_1(A)$  band in the Raman spectrum of the film into a low-frequency region in comparison with  $v_1(A) = 342 \text{ cm}^{-1}$  in a-GeS<sub>2</sub> [3] is supposed to be caused by the availability of internal deformations in the film at the initial growth stage. The availability of near-surface and transition film-substrate regions of structural units (s.u.) rich in germanium is also confirmed by X-ray photoelectronic investigations [3]. The central part of the films similar to bulk glass is formed presumably from  $GeS_{4/2}$  s.u. A joint consideration of Raman spectra structural analysis data at different depths of the  $\bar{E}^2$  maximum penetration into the film at  $\lambda_{\text{ex}} = 4800 \text{ Å}, \text{ and } \lambda_{\text{ex}} = 7500 \text{ Å} [3] (Fig. 2), the solution$ of back ellipsometric problem and the data of studying SIMS profile [3] allow us to propose the model of a layered-inhomogeneous film structure and its





Fig. 1 Depth dependence of the field intensity  $\overline{E}^2$  in  $\lambda/2(1)$ ,  $\lambda/4(2)$  films of the a-GeS<sub>2</sub>-type on a-GeS<sub>2</sub>-SiO<sub>2</sub>-Al three-layer structure



**Fig. 2** Depth dependence of IERS Raman spectrum  $As_x(GeS_2)_{100 \cdot x}$  films and glasses  $\lambda_{ex} = 6300 \text{ A}$ : 1 - x = 3, d = 800 Å; 2 - x = 3, d = 8000 Å;  $3\text{-}As_3(GeS_2)_{97}$ -glass;  $4\text{-}GeS_2$ -glass



Fig. 3 Model of the non-uniform layer structure for a-GeS<sub>2</sub>-type films (1) and model non-uniform profile of refractive index at  $\lambda_{ex} = 6300 \text{ Å}$  for a-GeS<sub>2</sub> (2)

refractive profile (Fig. 3). The availability of the nearsurface layer  $d_1$  (up to 50 Å) as well as the transition film-substrate layer  $d_2$  with a thickness up to 300 Å is common to thermally evacuated a-GeS<sub>2</sub>-type films.

## Conclusion

The structural interpretation of Raman spectra of a-GeS<sub>2</sub>-type films at different positions of the  $\overline{E}^2$  maximum in the layer together with the data of SIMS profiles and the depth dependence of refractive indices point to a layered-inhomogeneous distribution of the structure, composition and refractive index profile over the thickness of films prepared by flash evaporation.

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